# Photoelectrochemical removal of chlorfenvinphos by using WO<sub>3</sub> nanorods: influence of annealing temperature and operation pH

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#### ABSTRACT

A visible-light driven photoelectrochemical degradation process has been applied to a solution polluted with the organophosphate insecticide chlorfenvinphos. Different WO<sub>3</sub> nanosheets/nanorods have been used as photoanodes. These nanostructured electrodes have been fabricated by anodization of tungsten and, subsequently, they have been subjected to a thermal treatment (annealing). The combined influence of annealing temperature (400° C and 600° C) and operation pH (1 and 3) on the photoelectrocatalytic behavior of these nanorods has been examined through a statistical analysis. Morphological, structural and photoelectrochemical characterizations have also been carried out. The chlorfenvinphos degradation efficiency depended both on annealing temperature and, specially, operation pH. At pH 1 and using an annealing temperature of  $600^{\circ}$  C, chlorfenvinphos has been effectively degraded following pseudo-first order kinetics with a coefficient of  $7.8 \times 10^{-3}$  min<sup>-1</sup>, and notably mineralized (more than 65% of Total Organic Carbon decrease).

**Keywords:** Photoelectrochemical degradation; WO<sub>3</sub> nanorods; anodization; chlorfenvinphos; regression model.

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# **1. INTRODUCTION**

Organophosphate insecticides constitute a large group of synthetic organic compounds, highly toxic in many cases, whose precedent are the so-called "nerve gases" or "nerve agents" (sarin, tabun, soman, etc.) which were developed after World War II. The properties of these compounds made them very useful against insect plagues, so they have been used since then, as active ingredients, in many commercial formulations [1].

Organophosphate insecticides, such as parathion, diazinon or chlorfenvinphos, have been or are still widely used in agriculture, homes, gardens and veterinary practices. Once these insecticides have entered the organism of insects (and other animals, including human beings), cholinesterase-type enzymes, especially acetylcholinesterase, become their target molecules. This enzyme has a prominent physiological function in the nervous system. Transmission of nerve impulses from nerve fibers to smooth and skeletal muscle cells is inhibited by the reaction with organophosphate pesticides through phosphorylation, hence poisoning the organism [2]. Apart from their toxicity, these insecticides are also very persistent in the environment, with reported half-lives of 100-200 days in water [3].

Chlorfenvinphos –or 2-chloro-1-(2,4 dichlorophenyl) vinyl diethyl phosphate– is commonly used for protection against insects in agriculture and to control insect pests (flies, fleas, ticks, mites, etc.) on farm animals [4]. According to the European Chemical Agency, this insecticide is fatal if swallowed, toxic in contact with skin and very toxic to aquatic organisms with permanent effects [5]. This insecticide has been found in several aquifers (rivers, lagoons, ...), as well as in effluents from sewage treatment plants [3, 6-24]. Chlorfenvinphos can enter the environment from runoff due to rainfall, leaching from hazardous waste sites through large regions of soil and eventually reaching water resources, including underground wells [4], where it may threaten the aquatic environment and human health. The detection of this insecticide has been especially noticeable in the Mediterranean area, in spite of chlorfenvinphos being withdrawn from the EU [25].

The removal of hazardous pesticides from polluted surface waters and wastewaters cannot be efficiently performed by biological techniques, since they are toxic for microorganisms, thus hindering biodegradation. Other methods, based on the generation of the very oxidizing hydroxyl radicals, called Advanced Oxidation Processes (AOPs), have been used to remove chlorfenvinphos from aqueous solutions [10, 26-34]. However, most of the used methods involve high energies (gamma irradiation, low pressure UV photolysis) or require the use of expensive additional reagents (H<sub>2</sub>O<sub>2</sub>, O<sub>3</sub>, etc.) which make them not suitable to treat large volumes of water. Heterogeneous photocatalysis (PC) using semiconductor oxides could be an alternative, but the efficiencies of these processes are frequently limited by the fast recombination of photogenerated electron-hole pairs  $(e^{-}h^{+})$ , due to the absence of an electric field inside This problem can be overcome by using semiconductor [35, 36]. the photoelectrocatalysis (PEC), where an external low bias is applied to the photoelectrocatalyst to help separating  $e^-h^+$ , hence enhancing the overall efficiency [35]. To our knowledge, no photoelectrocatalytic process has been reported yet in the literature for chlorfenvinphos degradation. Tungsten trioxide (WO<sub>3</sub>) is a good candidate to be used as semiconductor photoanode in PEC due to its good properties, especially its

photoelectrocatalytic properties [37]. No published study where WO<sub>3</sub> was used as (photo)catalyst for chlorfenvinphos degradation has been found.

The aim of this study is to investigate the influence of two important parameters, annealing temperature and operation pH, on the photoelectrocatalytic performance of  $WO_3$  nanostructures, through a statistical analysis. Solutions of chlorfenvinphos were degraded using the WO<sub>3</sub> photoanodes under visible-light illumination. Before the degradation tests, these photoanodes were morphological, structural and electrochemically characterized through Field Emission Scanning Electronic Microscopy (FESEM), Raman spectroscopy and Electrochemical Impedance Spectroscopy (EIS).

## **2. EXPERIMENTAL PROCEDURE**

### 2.1. Nanostructures fabrication

Nanostructured WO<sub>3</sub> photoanodes were fabricated by anodization of tungsten cylinders (8 mm in diameter) under hydrodynamic conditions using a rotating disk electrode (RDE) at 375 rpm. Before anodization, tungsten cylinders were wet abraded with 220, 500 and 4000 grit SiC papers, rinsed with distilled water and dried with compressed air. The tunsgten cylinders were coated with teflon in order to expose only a circular area of  $0.5 \text{ cm}^2$  to the electrolyte. The anodization electrolyte was a  $1.5 \text{ M H}_2\text{SO}_4 + 0.05 \text{ M}$   $\text{H}_2\text{O}_2$  solution. A potential difference of 20 V was imposed between the anode (tungsten) and the cathode (platinum mesh). The temperature of the electrolyte (50° C) was controlled by a thermostatic bath.

After anodization, thin slices containing the formed nanostructures were cut from the tungsten cylinders and subjected to a thermal treatment for 4 h using two different annealing temperatures: 400° C and 600° C, at a heating rate of 15° C min<sup>-1</sup>. This thermal treatment was performed in order to obtain dehydrated and crystalline WO<sub>3</sub> nanostructures.

#### 2.2. Sample characterization

Morphology of WO<sub>3</sub> nanostructures was characterized through FESEM, using an acceleration voltage of 3 kV, while their composition and crystalline structure were studied through Raman spectroscopy, using a Raman Confocal Laser Microscope with a neon laser wavelength of 632 nm and a power of 420  $\mu$ W. Energy Dispersive X-ray spectroscopy (EDX) was also used to verify the chemical composition of the WO<sub>3</sub> nanostructures, applying an accelerating voltage of 20 kV. Finally, EIS experiments were performed under illumination conditions (using 420 nm light) in order to investigate the photoelectroactivity of the samples. For EIS measurements, a three-electrode transparent quartz cell was used, with the nanostructures acting as working electrodes, a platinum tip acting as a counter electrode and a Ag/AgCl (3M KCl) reference electrode. An Autolab PGSTAT302N potentiostat was used to apply an external polarization of 1 V and to record the impedance values. The frequency range varied from 10 kHz to 10 mHz, using a sinusoidal potential perturbation of 10 mV peak to peak.

## 2.3. Photoelectrocatalytic degradation experiments

The pesticide used in this study was chlorfenvinphos (SigmaAldrich Pestanal, analytical standard). Chlorfenvinphos solutions were prepared from a stock solution of 100 ppm, which was stored at 4° C in the absence of light. Degradation tests were performed at 20° C using the same electrode configuration explained above for the EIS experiments, applying an external potential difference of 1 V to the system and under constant magnetic stirring. The initial concentration of chlorfenvinphos was 20 ppm and the treated volume was 14 mL. Two different operation pH values were used: pH 1 and pH 3, by adding the necessary amount of H<sub>2</sub>SO<sub>4</sub>. Irradiation was provided by a 1000 W Xe light source, selecting a visible wavelength of 420 nm. The light power density was 100 mW cm<sup>-2</sup>.

## 2.4. Analytical methods

The pesticide concentration variation with time was monitored by UV spectrophotometry, measuring the absorbance within a range of wavelengths (190-320 nm). Liquid samples of 4 mL were withdrawn from the quartz reactor every 30 min to measure their absorbance. After measurements, these samples were returned to the reactor to continue with the degradation. The degradation tests lasted 360 min.

Mineralization of chlorfenvinphos after 360 min of degradation was monitored by measuring the Total Organic Carbon (TOC) of the treated solutions using a Shimadzu TOC-L analyzer.

# **3. RESULTS AND DISCUSSION**

#### 3.1. Nanostructures characterization

As explained above in the experimental section, WO<sub>3</sub> nanostructures were exposed to thermal treatment for 4 h in order to obtain dehydrated and crystalline structures. To study the influence of annealing temperature on the photoelectrocatalytic behavior of the nanostructures, two different values were selected: 400° C and 600° C.

The morphological characterization of the nanostructures was performed through FESEM. Figure 1a and Figure 1b show different images of the formed nanostructures after the annealing treatment. It can be observed that in both cases, very small nanorods or nanosheets were formed on the electrode surface forming a kind of spongy layer. The very small size of these nanostructures and the way they aggregate can be associated with the tendency of the bidentate  $O_2^{2^2}$  ligand to form very strong bonds with tungsten during anodization, which made the growth of the nanostructures difficult during their synthesis [38]. However, annealing temperature had an important influence on the nanostructures morphology. At 600° C, nanorods were much more defined, probably implying a higher degree of dehydration of the nanostructure. The cross sections of the nanostructures are shown as insets in Figure 1a and Figure 1b. It can be seen that these minuscule nanorods or nanosheets aggregated in many layers. From different cross-sectional images, the thickness of the nanostructured layer was determined to be  $2.1 \pm 0.3 \ \mu m$  for the nanostructure annealed at 400° C, and  $1.7 \pm 0.2 \ \mu m$  at 600° C. This

slightly decrease in thickness at the highest temperature may also be a consequence of a higher degree of dehydration, resulting in a shrinking of nanorods.

A current density transient recorded during the 4 h of anodization is presented in **Figure 1c**, with the aim of studying the formation mechanism of the WO<sub>3</sub> nanostructures used in this work. Just after applying the 20 V of cell potential difference, a steep decrease in current density was observed. This decrease was caused by the formation of a compact WO<sub>3</sub> layer on the exposed surface, which blocked the tungsten surface, according to the following reaction:

$$W + 3H_2O \rightarrow WO_3 + 6H^+ + 6e^-$$
(1)

After that decrease, current density started to increase until about 2500 s, which was related to the dissolution of the aforementioned compact WO<sub>3</sub> layer due to the interaction between the peroxo ligand ( $O_2^{2^-}$ ) and WO<sub>3</sub>. This reaction resulted in the formation of peroxotungstates and peroxotungstic acids, of which the species [ $W_2O_{11}$ ]<sup>2-</sup> seems to be predominant [38]:

$$2WO_3 + 4H_2O_2 \rightarrow [W_2O_{11}]^2 - + 2H^+ + 3H_2O$$
 (2)

When supersaturation conditions were achieved near the electrode due to the continuous release of peroxotungstates and peroxotungstic acids, these species precipitated on the tungsten surface in the form of tungstic acids (WO<sub>3</sub>·H<sub>2</sub>O and WO<sub>3</sub>·2H<sub>2</sub>O). Besides,  $W_2O_{11}^{2-}$  can also decompose into solid WO<sub>3</sub>, according to:

$$[W_2O_{11}]^{2-} + 2H^+ \rightarrow 2WO_3 + 2O_2 + H_2O$$
(3)

Precipitation of tungstic acids and  $WO_3$  in the form of nanostructures resulted in a constant decrease of current density, which reached low values at the end of the 4 h of anodization, similar to those observed at the beginning of the process.

In order to confirm the higher dehydration degree experimented by the nanostructure annealed at 600° C, Raman spectroscopy was used. Figure 2a shows Raman spectra of the WO<sub>3</sub> nanostructures after annealing at 400° C and 600° C. Characteristic monoclinic WO<sub>3</sub> peaks were clearly seen in both spectra (190 cm<sup>-1</sup>, 275 cm<sup>-1</sup>, 715 cm<sup>-1</sup> and 807 cm<sup>-1</sup>) [38-43]. However, there are two evident differences between both plots. For the sample annealed at 400° C, a broad and low-intensity band centered at ~950 cm<sup>-1</sup> can be observed, while this band is absent for the sample annealed at 600° C. This band has been traditionally associated with amorphous WO<sub>3</sub>·xH<sub>2</sub>O [40-44]. This result indicate that the nanostructure treated at 400° C was somewhat hydrated whereas at 600° C dehydration was complete. Moreover, a peak at 330 cm<sup>-1</sup> appeared for the sample annealed at 600° C, while this peak was barely visible for the sample treated at 400° C. This peak has also been related to crystalline monoclinic WO<sub>3</sub> [39-43]. Consequently, it can be concluded that the degrees of crystallinity and dehydration of the WO<sub>3</sub> nanostructure annealed at 600° C were higher than those for the sample annealed at 400° C. In Figure 2b, an EDX spectrum of the WO<sub>3</sub> nanostructure annealed at 400° is presented by way of illustration. Peaks of tungsten and oxygen can clearly be seen, confirming that WO<sub>3</sub> was formed.

With the aim of investigating the influence of annealing temperature on the photoelectrochemical activity of the nanostructures, EIS measurements under a 420 nm

light were performed. Nyquist and Bode-phase plots for the WO<sub>3</sub> nanostructures are shown in Figure 3. Two semicircles can be observed in each Nyquist plot (Figure 3a), one at high frequencies and the other at intermediate and low frequencies. The amplitude of these semicircles was visibly lower for the sample annealed at 600° C, which means that charge transfer processes under illumination were, in general, favored at this nanostructure (lower charge transfer resistances). Bode phase plots (Figure 3b) displayed two clear peaks (400° C) or one broad peak (600° C), which can be the result of two overlapping peaks. These results indicate that, at least, an electric equivalent circuit with two time constants should be used to fit the EIS results. To quantify charge transfer resistances, the electric equivalent circuit shown in Figure 3c has been used. In this circuit,  $R_S$  represents the electrolyte resistance, the  $R_1$ -CPE<sub>1</sub> time constant is related to charge-transfer processes at the nanostructure/electrolye interface and the  $R_2$ -CPE<sub>2</sub> time constant is related to a compact WO<sub>3</sub> layer formed beneath the nanorods. Constant phase elements (CPEs) have been used instead of capacitors to take the non-ideality of the system into account. Charge-transfer resistance values are shown in Table 1. Both  $R_1$  and  $R_2$  were lower for the nanostructure annealed at 600° C, confirming that chargetransfer processes were enhanced for this nanostructure under illumination. These results can be directly associated with the higher degree of crystallinity of this nanostructure, which improved electrochemical and photoelectrochemical interfacial phenomena.

#### 3.2. Photoelectrocatalytic degradation of chlorfenvinphos and mineralization degree

Apart from annealing temperature (400° C and 600° C), the operation pH was also modified between two levels: pH 1 and pH 3. Figure 4 shows the temporal evolution of chlorfenvinphos UV absorbance spectra for the different experimental conditions. The initial chlorfenvinphos spectrum showed two clear absorbance peaks centered at 205 nm and 245 nm. The second peak (245 nm) has been related in the literature to the  $\pi$ - $\pi$ \* transition of the aromatic ring and it is the peak normally used to track the decomposition of this organophosphate pesticide [32, 45]. It can be observed from Figure 4 that the intensity of both peaks decreased with irradiation time, regardless of the experimental conditions. However, this decrease was much more evident for the experiments carried out at pH 1. After 360 min of degradation and at pH 1, the absorbance peak centered at 245 nm completely disappeared for the nanostructure annealed at 600° C and took values close to zero for the nanostructure annealed at 400° C (but was not totally absent). Moreover, this peak was not displaced to lower wavelengths during the degradation reaction, indicating that intermediate compounds such as 2,4-dichlorophenol or 2,4-dichlorobenzoic acid were not formed. The formation of these compounds would imply a red-shift of this peak [32]. On the other hand, absorbance between 200-220 nm took values higher than zero. These results mean that (1) degradation of chlorfenvinphos likely took place through the aromatic ring opening, and (2) this degradation resulted in the formation of other intermediate compounds whose molecules absorbed light in the wavelength range between 200 nm and 220 nm. Nevertheless, further analytical work is necessary to propose a complete degradation mechanism for chlorfenvinphos. This study will be developed in future works.

The decrease of the absorbance at 245 nm was related to the concentration of chlofenvinphos by using a calibration line. The degradation efficiency was determined according to the following equation:

$$\eta = \frac{C_0 - C(t)}{C_0} \times 100$$
 (1)

where  $C_0$  is the initial chlorfenvinphos concentration and C(t) is chlorfenvinphos concentration at different times. Degradation efficiencies are shown in **Figure 5a**. At pH 3, little differences were observed between WO<sub>3</sub> samples, and degradation efficiencies of 37% (600° C) and 34% (400° C) were obtained. At pH 1, although differences were not marked, the WO<sub>3</sub> nanostructure annealed at 600° C displayed the best degradation efficiency (95% vs. 91% for the nanostructure annealed at 400° C).

Pseudo-first order kinetic coefficients for chlorfenvinphos degradation are presented in **Table 2**, as well as the respective correlation coefficients ( $R^2$ ) for the linearization. At 600° C, higher kinetic coefficients were obtained, regardless of the operation pH. On the other hand, at pH 1, these values were much higher, especially at 600° C, which is consistent with the previous results. As stated in the introduction, this is the first study involving photoelectrocatalytic degradation of chlorfenvinphos. Therefore, kinetic coefficients obtained here cannot be compared with other coefficients obtained under similar experimental conditions. In order to make a rough comparison, a study about photocatalytic degradation of chlorfenvinphos by using suspensions of TiO<sub>2</sub> will be used [29]. In that work, 200 mg L<sup>-1</sup> of TiO<sub>2</sub> photocatalyst were used, in a solar reactor with a total volume between 72-80 L and using a solar collector composed of four compound parabolic collector units. The obtained pseudo-first order kinetic coefficient

for chlorfenvinphos degradation was 0.014 min<sup>-1</sup>, higher than the 0.0078 min<sup>-1</sup> obtained in this work. However, to adequately compare these kinetic coefficients, experimental conditions should be similar, and the solar pilot plant used by Malato et al. is very different from the laboratory degradation system used in this work.

Mineralization degree of chlorfenvinphos was determined through the analysis of total organic carbon. TOC values for all different conditions, together with the percentage of TOC decrease, are shown in **Figure 5b**. Initial values are also shown for comparison. At pH 3, the maximum TOC decrease was obtained for the nanostructure annealed at 600° C (19 %), while for the nanostructure annealed at 400° C, that reduction was of 16.9 %. At pH 1, TOC values significantly decreased, reaching a 65.3 % of TOC reduction for the nanostructure treated at 600° C (and 60.5 % for sample annealed at 400° C).

Consequently, in accordance with the previous results, it can be concluded that WO<sub>3</sub> nanostructures used in the present study are efficient for their use as photoanodes in the PEC degradation of chlorfenvinphos.

# 3.3. Regression analysis: influence of annealing temperature and operation pH

With the purpose of finding the relationship between the two factors studied in this work (annealing temperature *T* and operation pH) on the photoelectrocatalytic behavior of WO<sub>3</sub> nanostructured photoanodes, a regression analysis was performed. The kinetic coefficient (*k*) was used as response variable and the purpose was to maximize it. A  $2^2$  factorial design was selected, since it combines simplicity and effectiveness. The equation of the regression model is:

$$k = \beta_0 + \beta_1 \cdot T + \beta_2 \cdot pH + \beta_3 \cdot T \cdot pH$$
<sup>(2)</sup>

According to ANOVA, a statistically significant result is obtained when the *p*-value is lower than the stipulated significance level. In this case, this upper limit was set at 0.05. Therefore, for a given factor, *p*-values lower than 0.05 mean that this factor had a significant influence on the response variable, an effect that cannot be explained by simple random chance. In **Table 3**, *p*-values for each factor and for the interaction between them, as well as the regression coefficients  $\beta_0$ ,  $\beta_1$ ,  $\beta_2$  and  $\beta_3$ , are shown (the parameter  $\beta_0$ , which is the expected value of *k* when *T* and *pH* are zero, has no physical meaning, but its value is also shown in Table 3 to provide a complete version of the regression model). It can be seen that all *p*-values are lower than 10<sup>-4</sup>, which indicates that the effects of both factors and the interaction were significant.

The main effects and interaction plots are shown in **Figure 6**. Lines observed in the main effects plot (**Figure 6a**) indicate the estimated changes in the kinetic coefficient when each factor is moved from its lowest level to its highest level, setting the other factor at an intermediate value between the lowest and highest level (i.e. setting  $T = 500^{\circ}$  C and pH = 2). It is clear that the factor with the most significant effect was operation pH, which has to be set at its lowest limit (pH 1) in order to maximize the kinetic coefficient (**Figure 6a**). The influence of annealing temperature was smaller, although it was also found to be significant: to maximize *k*, annealing temperature must be set at its maximum value (600° C) (**Figure 6a**). Concerning the interaction between both factors, its effect was also significant, since no parallel lines are observed in

**Figure 6b**. From these results, it can be concluded that optimum values are pH 1 and 600° C.

The strong influence of operation pH can be explained by taking into account the chemistry of tungsten species. WO<sub>3</sub> is chemically stable in acidic solutions. In fact, precipitation of hydrated forms of WO<sub>3</sub> (known as tungstic acids), which was the formation mechanism of nanostructures used in this work, is achieved at pH of 1 or lower [46]. Hence, a better photoelectrochemical performance of WO<sub>3</sub> photoelectrodes is expected in acidic environments. On the other hand, the higher degrees of dehydration and crystallinity of nanostructures annealed at 600° C (see **Figure 2**) would explain the enhancement of degradation efficiency of these samples in comparison with samples treated at 400° C.

# 4. CONCLUSIONS

Very small nanorods or nanosheets were formed by electrochemical anodization of tungsten, regardless of the annealing temperature. Nevertheless, at 600° C, nanorods were much more defined due to a higher degree of dehydration and crystallinity of the nanostructure. Hence, annealing temperature had an important influence on the nanostructures morphology.

The higher degree of dehydration and crystallinity of nanorods obtained at 600° C was confirmed by Raman spectroscopy. A peak directly related to amorphous and hydrated

WO<sub>3</sub> was observed for the sample annealed at 400° C, but not for the sample annealed at  $600^{\circ}$  C.

Charge-transfer resistances under illumination were lower for the nanostructure fabricated at 600° C. This result has been associated with the higher degree of crystallinity of this nanostructure, which improved electrochemical and photoelectrochemical interfacial phenomena.

The UV absorbance peak associated with the aromatic ring of chlorfenvinphos (at 245 nm) decreased with irradiation time for all experimental cases. However, this decrease was much more pronounced at pH 1 than at pH 3. The temporal evolution of this peak indicated that degradation likely proceeded through the cleavage of the aromatic ring, rather than through the formation of intermediate compounds such as 2,4-dichlorophenol or 2,4-dichlorobenzoic acid, although future investigations are necessary to confirm this preliminary results.

More than 65% of the initial chlorfenvinphos was mineralized after 360 min of degradation reaction for the nanostructure annealed at 600° C and under an operation pH of 1.

A statistical analysis has shown that both annealing temperature and operation pH, as well as their interaction, had a significant effect on the pseudo-first order kinetic coefficient, especially the operation pH. In order to maximize this coefficient, annealing temperature must be fixed at its highest value (600° C) and pH at its lowest value (pH 1). The important influence of operation pH can be explained by taking into account the chemical and photoelectrochemical stability of WO<sub>3</sub> in highly acidic environments, which enhanced the photoelectrochemical performance of WO<sub>3</sub> photoelectrodes. On the other hand, the higher degrees of dehydration and crystallinity of nanostructures annealed at  $600^{\circ}$  C explained the improvement of degradation efficiency of these samples in comparison with samples treated at  $400^{\circ}$  C.

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# **Tables captions**

**Table 1**. Charge-transfer resistance values ( $R_1$  and  $R_2$ ) for the WO<sub>3</sub> nanostructures.

**Table 2**.Pseudo-first order kinetic coefficients for chlorfenvinphos degradation (*k*) for the different experimental conditions.

**Table 3.** *p*-values for each factor (A: Temperature; B: pH) and for the interaction between them (AB). The regression coefficients defined in equation (2) as  $\beta_0$ ,  $\beta_1$ ,  $\beta_2$  and  $\beta_3$  are also shown.

#### **Figures captions**

**Figure 1.** FESEM images (at 15 KX) of the formed nanostructures after the annealing treatment at different temperatures: (a) 400° C, (b) 600° C; current density transient recorded during tungsten anodization (c).

**Figure 2.** (a) Raman spectra of the WO<sub>3</sub> nanostructures after annealing at 400° C and 600° C; (b) EDX spectrum of the WO<sub>3</sub> nanostructure after annealing at 400° C.

**Figure 3.** EIS experimental results for the WO<sub>3</sub> nanostructures presented in the form of Nyquist (a) and Bode-phase (b) plots. The equivalent circuit used to fit these results is also shown (c).

**Figure 4.** Temporal evolution of chlorfenvinphos UV absorbance spectra for the different experimental conditions.

**Figure 5.** (a) Degradation efficiencies for the different experimental conditions; (b) TOC values for all different conditions, together with the percentage of TOC decrease.

Figure 6. Main effects (a) and interaction plots (b) for kinetic coefficient k.

# Figure 1





Figure 2





Figure 3











400 600 **Temperature/° C** 

# Table 1

WO3 sample	$R_1/\Omega \ \mathrm{cm}^2$	$R_2/\Omega \ \mathrm{cm}^2$
400° C	$109 \pm 8$	$991 \pm 34$
600° C	$42 \pm 7$	$378 \pm 16$

# Table 2

Temperature/º C	pН	<i>k</i> /min <sup>-1</sup>	<i>k/s</i> <sup>-1</sup> (x10 <sup>-5</sup> )	$R^2$
400	3	0.0012	2.0	0.989
400	1	0.0062	10.3	0.989
600	3	0.0013	2.2	0.988
600	1	0.0078	13	0.971

# Table 3

Effect	<i>p</i> -value	Regression coefficient (x10 <sup>-5</sup> ; k in
		min <sup>-1</sup> )
		$\beta_0 = 400$
A: Temperature	<10 <sup>-4</sup>	$\dot{\beta}_{I} = 1.175$
B: pH	<10-4	$\beta_2 = -100$
AB	<10-4	$\beta_3 = -0.375$